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Note

Diblock copolymers of ethylene oxide and 1,2-butylene oxide in aqueous solution Formation of unimolecular micelles

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1. Introduction

Aqueous micellar solutions of block copolymers based on poly(ethylene oxide) as the hydrophilic component combined with a number of hydrophobic polymers have potential as vehicles for drug solubilisation, see, for example, reviews by[Attwood and Booth](#page-3-0) [\(2007\),](#page-3-0) [Chiappetta and Sosnik \(2007\),](#page-3-0) [Savic et al. \(2006\),](#page-3-0) [Gaucher](#page-3-0) [et al. \(2005\)](#page-3-0) and [Adams et al. \(2003\).](#page-3-0) A particular advantage of this family of copolymers is the so-called 'stealth' property of the poly(ethylene oxide) corona of their micelles which allows the drug-loaded micelles to evade scavenging by the mononuclear phagocyte system, so resulting in increased circulation times in the blood. In this note we focus attention on diblock copolymers prepared by sequential oxyanionic polymerisation of ethylene oxide followed by 1,2-butylene oxide. To describe the repeat units of the blocks we use the notation: $E = OCH₂CH₂$ (from ethylene oxide) and $B = OCH₂CH(C₂H₅)$ (from 1,2-butylene oxide), while subscripts *m* and *n* are used to denote number-average lengths in repeat units of the hydrophilic and hydrophobic blocks. For example, a diblock copolymer formed by sequential copolymerisation of ethylene oxide followed by 1,2-butylene oxide is denoted E*m*B*n*.

ABSTRACT

The dependence of log(cmc) on hydrophobic block length *n* was examined for E_mB_n copolymers (E = oxyethylene, B = oxybutylene, subscripts denote number-average block lengths in repeat units) with *n* in the range 30–76. Combination with published data for E*m*B*ⁿ* diblock copolymers with shorter E-blocks shows two changes of slope in the log(cmc)–*n* plot corresponding to the onset of unimolecular micelle formation at *n* ≈ 12 and completion of this process at *n* ≈ 30. The results are discussed with reference to published data for $E_m L_n$ and $E_m CL_n$ (L from p, L -lactide; CL from ε -caprolactone) copolymers, which show similar behaviour.

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As discussed previously (see, e.g., [Attwood et al., 2007; Booth et](#page-3-0) [al., 2006\),](#page-3-0) and for micelle association number $N \geq 50$ [\(Hall, 1987\),](#page-3-0) the standard Gibbs energy of micellisation is obtained without significant error from the critical micelle concentration (cmc) through

$\Delta_{\text{mic}}G^{\circ} = -RT \ln K_c = RT \ln(\text{cmc})$

where the cmc is expressed in mol dm−3, *K*^c is the unimer–micelle equilibrium constant, and the standard state is ideally dilute solution in which both unimers and micelles are of unit molarity. That is, log(cmc) is directly related to the standard Gibbs energy of micellisation at a given temperature, and can be used as a convenient indicator of the position of equilibrium in the system. If log(cmc/mol dm−3) is known for series of copolymers with the same hydrophilic component, then the relative hydrophobicity per chain unit can be readily extracted: see, for example, [Booth et al.](#page-3-0) [\(2006\)](#page-3-0) and [Attwood et al. \(2007\).](#page-3-0)

A change in the dependence of log(cmc) on hydrophobic block length indicates a change in the micellisation equilibrium. Such a change will occur if the dispersed copolymer molecules (unimers) start to form unimolecular micelles, i.e., when the longest coiled hydrophobic blocks in the distribution collapse to a globule, much as pictured by [Brown et al. \(1989\),](#page-3-0) [Tuzar and Kratochvil](#page-3-0) [\(1993\),](#page-3-0) [Chu \(1995\)](#page-3-0) and [Cooke and Williams \(2003\).](#page-3-0) The consequence of collapse of the hydrophobic block is reduced contact of the chain units of the core-forming block with water and so

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a reduction in the hydrophobic effect which drives micellisation, as discussed in detail elsewhere [\(Kelarakis et al., 2001\).](#page-3-0) Because of the block-length distribution, the conversion of unimer to unimolecular micelle changes gradually as the average value of *n* is increased, and another change in the dependence of log(cmc) on hydrophobic block length is expected at higher values of *n* when effectively all dispersed molecules are in the form of unimolecular micelles. As reviewed recently ([Attwood et al., 2007\),](#page-3-0) published values of the cmc for poly(ethylene oxide)/poly(p,L-lactide) and poly(ethylene oxide)/poly(&-caprolactone) diblock copolymers (denoted $E_m L_n$ and $E_m CL_n$ respectively, where $L = COOCH(CH_3)$ and $CL = COO(CH₂)₅$) cover a wide range of hydrophobic block length ($n = 12-108$ for $E_m L_n$ copolymers and $n = 2-74$ for $E_m CL_n$ copolymers), and plots of log(cmc) against *n* for these two copolymers do indeed show two changes in slope.

Our interest in the formation of unimolecular micelles is not primarily in their solubilisation potential, although their hydrophobic interior may provide a site for the limited incorporation of waterinsoluble drugs, but in their effect on the dependence of the value of the cmc on hydrophobic-block length for all types of micellisable block copolymers. The first change in slope has been reported for certain poly(ethylene oxide)/polyether diblock copolymers ([Booth](#page-3-0) [et al., 2006\)](#page-3-0) but, because the ranges of block lengths available were narrow, the second transition was not seen. For example, apart from a recently published result for $E_{45}B_{26}$ ([Elsabahy et al.,](#page-3-0) [2007\),](#page-3-0) values of the cmc for E*m*B*ⁿ* copolymers have been restricted to relatively short block lengths, *n* = 7–18. In this note we report new determinations for E*m*B*ⁿ* copolymers with *n* in the range 30–76.

2. Experimental

Five E*m*B*ⁿ* copolymers with narrow chain-length distributions $(M_w/M_n$ ≈ 1.05) and long B-block lengths ($E_{110}B_{30}$, $E_{209}B_{45}$, $E_{100}B_{51}$, $E_{114}B_{56}$, and $E_{155}B_{76}$) were available from a previous study: see [Ryan et al. \(2001\)](#page-3-0) for details. 1,6-Diphenyl-1,3,5-hexatriene (DPH) was obtained from BioChemika (Fluka) and used as received. Stock copolymer solutions were prepared by dissolving the copolymers in Milli-Q water, allowing 24 h for complete dissolution before diluting to required concentrations within the range $0.01-14$ mg dm^{−3}. Solubilisation of DPH was used to determine the onset of micellisation, as in an investigation of triblock copolyethers by [Alexandridis](#page-3-0) [et al. \(1994\),](#page-3-0) and before that an investigation of ionic surfactants by [Chattopadhyay and London \(1984\). D](#page-3-0)PH was dissolved in methanol and added to the copolymer solution, so that the final copolymer solution contained 1% (v/v) methanol and 0.004 mM DPH, a mixture shown to provide the same values of the cmc as those obtained by other methods for copolymers in water alone. An F-4500 Hitachi fluorescence spectrophotometer was used in the experiments, with solution temperatures maintained at 25 or 30 ± 0.2 $^{\circ}$ C.

3. Results

Preliminary investigations indicated advantage in using the intensity of fluorescence of DPH at 430 nm rather than the intensity of absorption at 356 nm. Examples of the dependence of fluorescence intensity on copolymer concentration (logarithmic scale) are illustrated in Fig. 1. The value of the cmc was obtained from the intersection of the straight line through the data points with the baseline. Values of the cmc at 25 ◦C obtained for the five copolymers are listed in Table 1, together with the molar masses of the copolymers taken from [Ryan et al. \(2001\). S](#page-3-0)olutions of two of the copolymers, $E_{110}B_{30}$ and $E_{209}B_{45}$, were also studied at 30 °C: values of the cmc were unchanged.

Fig. 1. Dependence of the intensity of fluorescence on the logarithm of copolymer concentration in aqueous solutions of copolymers $E_{209}B_{45}$ and $E_{155}B_{76}$, as indicated.

Table 1

Number-average molar masses and critical micelle concentrations (*T* = 25 ◦C) of E*m*B*ⁿ* copolymers

Copolymer	M_n (g mol ⁻¹)	cmc (mg dm ⁻³)
$E_{110}B_{30}$	7,000	0.50
$E_{209}B_{45}$	12,400	1.70
$E_{100}B_{51}$	8,070	0.30
$E_{114}B_{56}$	9,050	0.37
$E_{155}B_{76}$	12,300	0.40

In passing we note that poly(ethylene oxide)-based copolymers with long hydrophobic blocks and high hydrophobic/hydrophilic ratios are known to form vesicles in aqueous solution. This includes E*m*B*ⁿ* copolymers of the type under discussion (see, e.g., [Harris et](#page-3-0) [al., 2002; Kelarakis et al., 2008\).](#page-3-0) However, vesicles do not form at the low concentrations typical at the cmc, e.g. >0.002 g dm^{-3} , see Table 1.

4. Discussion

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Combining the values of the cmc in Table 1 with previously published values (summarised in Table 2) gives the plot of log(cmc/mol dm−3) against B-block length shown in [Fig. 2.](#page-2-0) Variation in the temperature of determination of the cmc (see Table 1) is not a problem since, as discussed below, values of

T = 30 °C: exception Elsabahy et al., 20 °C.

Fig. 2. Dependence of log(cmc) on the length of the B block (*n*) for block copolymers of poly(ethylene oxide) and poly(1,2-butylene oxide): $\left(\bullet \right)$ published values $($ see [Table 2\),](#page-1-0) $($ $\blacksquare)$ this work.

the cmc of E*m*B*ⁿ* copolymers are insensitive to temperature when $n \geq 15$ units. Variation in the E-block length has a significant effect, and in constructing Fig. 2 we used d $log_{10}(cmc)/dm = 0.004$ to adjust the values of the cmc (molar units) to a common number Eblock length *m* = 100, a procedure based on results published by [Alexandridis et al. \(1994\)](#page-3-0) and discussed previously [\(Booth et al.,](#page-3-0) [2006\).](#page-3-0) As anticipated, the data points can be satisfactorily represented by lines showing two transitions, consistent with formation of unimolecular-micelles starting at $n \approx 12$ and completion of the process at $n \approx 30$.

The temperature dependence of log(cmc) gives a value of the van't Hoff enthalpy of micellisation from

$$
\Delta_{\text{mic}} H_{\text{VH}} = \frac{R \, \text{d} \ln(\text{cmc})}{\text{d} (1/T)}
$$

For large values of the micelle association number ($N \geq 50$, [Hall,](#page-3-0) [1987\)](#page-3-0) $\Delta_{\text{mic}}H_{\text{VH}}$ is a true value of the standard enthalpy of micellisation, the usual situation when *n* is large. For smaller values it is to be regarded as an apparent value of the standard enthalpy which, nevertheless, correctly describes the temperature dependence of the cmc. Plotting values of the van't Hoff enthalpy per B unit ($\Delta_{\text{mic}}H_{\text{VH}}/n$) against block length brings the data for E_mB_n copolymers into acceptable correspondence; see Fig. 3. Very low values, essentially $\Delta_{\text{mic}}H_{\text{VH}}$ = 0, are characteristic of E_mB_n copolymers with $n \geq 15$, as might be expected for transfer of a copolymer molecule with a collapsed coil to a micelle core. As noted above, values of the cmc which are independent of temperature, i.e., consistent with a van't Hoff enthalpy of micellisation of zero, have been found for other block copolymers with long hydrophobic blocks (e.g. [Yamamoto et al., 2002\).](#page-3-0)

Present results for E*m*B*ⁿ* copolymers are compared with the results for E*m*L*ⁿ* and E*m*CL*ⁿ* copolymers in Fig. 4 where, for clarity, they are represented by the lines drawn through the data points. The results for E*m*L*ⁿ* and E*m*CL*ⁿ* copolymers are taken directly from the review by [Attwood et al. \(2007\), w](#page-3-0)here they are corrected to a common E-block length, *m* = 100. As described previously ([Attwood](#page-3-0) [et al., 2007\),](#page-3-0) for values of *n* below that required for formation of unimolecular micelles, effectively for log(cmc/mol dm⁻³) > -5, superposition of results by scaling *n* leads to the following ranking

Fig. 3. Dependence of the van't Hoff enthalpy of micellisation per hydrophobic unit on block length for E*m*B*ⁿ* copolymers. The curve is intended to lead the eye through the data. \bullet results from [Bedells et al. \(1993\),](#page-3-0) [Tanodekaew et al. \(1993\),](#page-3-0) [Yu et al.](#page-3-0) (1997) , Kelarakis et al. $(1998, 2002)$ and Chaibundit et al. (2002) , (\blacksquare) present work.

of hydrophobicities per chain unit:

 $L : B : CL = 1 : 1.5 : 2$

If the effect of unimolecular–micelle formation on cmc depended only on hydrophobicity (defined in this way) then the level of the lines at high values of *n* should fall in order $E_m L_n$ > $E_m B_n$ > $E_m C L_n$. While the results for the two copolymers with polyester blocks fit this pattern, that for E*m*B*ⁿ* copolymers does not.

Accordingly, we seek an explanation which takes account of the chemical difference between the polyether and polyester blocks. Small-angle neutron scattering (SANS) has been used to show that the micelle cores of $E_m B_n$ copolymers contain water, an effect ascribed to association with the ether oxygens of the B-block and also to its terminal hydroxyl group ([Derici et al., 1999; Castelletto et](#page-3-0) [al., 2002, 2004\).](#page-3-0) The hydrophobic blocks of E*m*L*ⁿ* and E*m*CL*ⁿ* copoly-

Fig. 4. Dependence of log(cmc) on hydrophobic-block length for the length for E*m*L*n*, E*m*CL*ⁿ* and E*m*B*ⁿ* copolymers, as indicated. The lines summarise the results displayed in [Fig. 1](#page-1-0) and in [Attwood et al. \(2007\).](#page-3-0)

mers have ester oxygens, and terminal hydroxyl groups, and SANS has been used to study micellar solutions (Vangete et al., 2004; Riley et al., 2003), but not with a view to detection of water in the micelle core. As it happens, methylation of the hydroxy-end group of E*m*B*ⁿ* copolymers is known to make little or no difference to values of the cmc of copolymers with short block lengths $(E_{18}B_{10}$ and $E_{11}B_8$) forming large micelles ($N > 60$ at 25 °C) (Kelarakis et al., 2002; Chaibundit et al., 2002). However, the formation of unimolecular micelles (*N* = 1) is likely to be particularly affected by inclusion of water molecules in the collapsed coil, including water binding to the oxygens of the ether or ester linkages, and the seemingly anomalous result for the E*m*B*ⁿ* copolymers may well be a consequence of a differences in this interaction.

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